MEASUREMENT OF SYNCHROTRON PULSE DURATIONS USING SURFACE PHOTOVOLTAGE TRANSIENTS

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Abstract

We report results on experiments using combined laser and synchrotron radiation. Picosecond laser pulses at 800 nm are used to induce surface photovoltage transients in p-type Si samples. A two-component decay is observed. The fast component of decay provides a direct measure of synchrotron soft x-ray pulse durations.

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Processes driven by combined synchrotron (x-ray) and laser (optical) radiation provide a basis for novel scientific directions and technological applications of synchrotron light. From a fundamental perspective, synchrotron x-rays can probe transient states of matter initiated by laser excitation. From an applications perspective, x-ray interactions with a laser-perturbed material may be used to characterize synchrotron x-ray pulses. The surface photovoltage effect has been studied previously [1,2] and provides an opportunity for such synchrotron x-ray pulse characterization. We describe time-resolved measurements of Si 2p photoemission peak shifts induced by surface photovoltage transients. The photoemission peak dynamics provide a direct measure of the temporal duration of soft x-ray pulses at the Advanced Light Source (ALS).

Experiments are performed at the ALS micro-XPS beamline (BL 7.3.1.2). The beamline provides a micro-focused x-ray spot (diameter < 5 μm) and typically delivers a flux of ~10^{10} γ/sec*0.1% BW. Soft x-ray photons at ~700 eV photoionize a p-type Si (111) sample and the x-ray photoelectron spectrum (XPS) is measured using a commercial hemispherical analyzer. Perturbations to the XPS are induced by a laser. The laser system is based on titanium-doped-sapphire and provides 800 nm pulses at a repetition rate of ~ 83 MHz (12 ns laser pulse spacing). The laser pulse energy is typically 1-2 nJ per pulse and the laser beam is focused to a 15 μm diameter spot resulting in an excitation fluence of ~1 mJ/cm². Laser pulses are temporally stretched in a fiber to a pulse duration of 5ps. The laser system is synchronized to the ALS storage ring (typically 1-2 ps RMS jitter) by driving a laser-cavity end-mirror with a piezoelectric actuator which in turn is driven by a phase-locked loop (whose inputs are an ALS RF signal and a laser photodiode signal).

The x-ray/laser spatio-temporal overlap is achieved as follows. Coarse spatial overlap is achieved by viewing a phosphor-coated region of the Si sample and overlapping x-ray and laser
beams on the phosphor. Fine spatial overlap is then obtained by translating the sample to a phosphor-free region and scanning the laser focal spot until a shift is observed in the Si 2p photoemission peak. We note that the Si 2p photovoltage shift does not fully relax between laser excitation pulses so that temporal overlap is not required to observe a Si 2p XPS shift (temporal overlap does of course maximize the shift). Using the above-described procedure and computer-controlled scanning of the laser focal spot (implemented using piezoelectric actuators and linear variable differential transformers), optimal overlap of the two sub-20 μm spots is easily and efficiently achieved.

Coarse temporal overlap is achieved using an avalanche photodiode (pulses overlapped typically to within 500 ps). The x-ray/laser time-delay is then optimized on a sub-500 ps timescale by scanning the time delay and maximizing the Si 2p peak shift. The x-ray/laser time delay is varied by shifting the phase of the ALS RF signal before it is sent into the phase-locked-loop. This method of varying the x-ray/laser time delay is attractive since it is purely electronic and physical motion of (laser) beam steering optics is avoided. Consequently the x-ray/laser spatial overlap is not compromised. In order to match the laser repetition rate, the Advanced Light Source is operated in a special mode whereupon every 6th bucket is filled (resulting in a 12 ns spacing between ALS pulses).

A laser induced shift to the Si 2p peak can be observed in Fig. 1 inset (dashed vs solid curves). One observes a laser-induced shift of ~450 meV to lower binding energy. Similar XPS shifts have been observed previously in Si and are due to a surface photovoltage [1]. In brief, the laser produces (photo-generated) carriers and these carriers move to offset band bending at the surface of the p-type Si sample. This carrier motion is rapid (typically sub-picosecond) and is unresolved in the present experiment (where time-resolution is set by the ~70 ps ALS pulse duration in this operating mode). This rapid carrier motion causes the XPS shift observed in Fig.
The data in the inset is obtained by recording spectra either with or without a laser pulse so that one either observes no shift or the full surface photovoltage shift. We have observed that the magnitude of the shift varies with surface quality. Smaller shifts (down to ~ 0 meV) are obtained if the Si surface is damaged from very high laser excitation fluence (in the J/cm² regime). Similarly the presence of surface contaminants or oxygen coverage can alter the magnitude of the shift.

Surface photovoltage dynamics are shown in Fig. 1 (main figure). In these measurements a laser always excites the Si and we measure the XPS shift (specifically electron counts over a spectral range from 101-102 eV) as a function of x-ray/laser relative arrival time at the sample. Under these conditions the XPS shift does not fully relax between laser excitation pulses. At short time-delay (< 350 ps) the x-ray probes the sample before laser excitation (here the x-ray sees a residual shift from the previous laser pulse). At a delay of ~350 ps the x-ray and laser pulse start to arrive simultaneously at the Si sample; a larger XPS shift is observed and one observes an increase in the electron count rate. The electron count rate will subsequently decrease as the XPS shift relaxes due to loss of photo-carriers (either from recombination, transport to the bulk...[1]).

The data of Fig. 1 indicates that there are two components which govern relaxation of the XPS shift. A slow component is observed beyond 500 ps; similar ‘slow’ dynamics have been observed previously [1] and are attributable to carrier transport within the space-charge layer of the semiconductor. A ‘fast’ component of decay is observed between 400 and 500 ps and is likely due to rapid electron-hole recombination at the semiconductor surface. Surface recombination is thought to be mediated by trap states (which lie within the semiconductor bandgap) and as such will depend on the nature of the surface[1-3]. Measurements of surface dynamics on annealed Si(111)2x1 surfaces indicate surface carrier decay times of order 100 ps
The samples used in the present experiments are Ar-ion sputtered Si(111)2×1 samples and the measurements of Fig. 1 indicate similar, though somewhat faster, carrier decay times. Somewhat faster decay times (compared to annealed surfaces) can result from increased density of trap/defect states associated with ion-sputtering.

The rapid material response associated with surface carrier dynamics gets convolved with the ALS pulse duration and as such provides a measure of the synchrotron pulse duration. We determine the width (full width at half-maximum) of this feature to be 70 ps.

In conclusion, we have used combined laser and synchrotron radiation to study the dynamics of surface photovoltage transients in Si. A fast component of photo-carrier decay permits a measure of the temporal duration of synchrotron x-ray pulses.

References

**Figure Caption**

**Figure 1.** Electron counts (over a 101-102 eV spectral range) from Si 2p photoemission peak vs x-ray/laser time-delay. At early delay (< 350 ps) the x-rays probe before laser excitation of the sample. When x-ray and laser pulses strike the Si sample simultaneously (starting at ~ 350 ps) an increased count rate is obtained due to a shift in the photoelectron peak. Partial relaxation of the shifted peak (occurring from ~400-500 ps) provides a direct measure of the ALS pulse duration (FWHM ~70 ps). The full Si 2p shift (~450 meV) is shown in the figure inset (dashed curve : no laser; solid curve = laser present).
Figure 1