Title
Unexpected Diffusion Anisotropy of Carbon Dioxide in the Metal-Organic Framework Zn2(dobpdc)

Permalink
https://escholarship.org/uc/item/3xd3z5rz

Journal
Journal of the American Chemical Society, 140(5)

ISSN
0002-7863

Authors
Forse, AC
Gonzalez, MI
Siegelman, RL
et al.

Publication Date
2018-02-07

DOI
10.1021/jacs.7b09453

Peer reviewed
Unexpected Diffusion Anisotropy of Carbon Dioxide in the Metal–Organic Framework Zn₂(dobpdc)

Alexander C. Forse,†,‡,§ Miguel I. Gonzalez, † Rebecca L. Siegelman, † Valenia J. Witherspoon, †‡ Sudi Jawahery, †§ Rocio Mercado, † Phillip J. Milner, † Jeffrey D. Martell, † Berend Smit, †‡,§,¶ Berhard Blümich, †,‡,¶,†‡,# and Jeffrey A. Reimer †,‡,§

†Department of Chemistry, ‡Department of Chemical and Biomolecular Engineering, and §Berkeley Energy and Climate Institute, University of California, Berkeley, California 94720, United States
Il Institut des Sciences et Ingenierie Chimiques, Valais, École Polytechnique Fédérale de Lausanne (EPFL), Rue de l’Industrie 17, CH-1951 Sion, Switzerland
† Institut für Technische und Makromolekulare Chemie (ITMC), RWTH Aachen University, 52062 Aachen, Germany
#Materials Sciences Division, Lawrence Berkeley National Laboratory, Berkeley, California 94720, United States

Supporting Information

ABSTRACT: Metal–organic frameworks are promising materials for energy-efficient gas separations, but little is known about the diffusion of adsorbates in materials featuring one-dimensional porosity at the nanoscale. An understanding of the interplay between framework structure and gas diffusion is crucial for the practical application of these materials as adsorbents or in mixed-matrix membranes, since the rate of gas diffusion within the adsorbent pores impacts the required size (and therefore cost) of the adsorbent column or membrane. Here, we investigate the diffusion of CO₂ within the pores of Zn₂(dobpdc) (dobpdc = 4,4'-dioxidobiphenyl-3,3'-dicarboxylate) using pulsed field gradient (PFG) nuclear magnetic resonance (NMR) spectroscopy and molecular dynamics (MD) simulations. The residual chemical shift anisotropy for pore-confined CO₂ allows PFG NMR measurements of self-diffusion in different crystallographic directions, and our analysis of the entire NMR line shape as a function of the applied field gradient provides a precise determination of the self-diffusion coefficients. In addition to observing CO₂ diffusion through the channels parallel to the crystallographic c axis (self-diffusion coefficient Dₜ = (5.8 ± 0.1) × 10⁻⁹ m² s⁻¹ at a pressure of 625 mbar CO₂), we unexpectedly find that CO₂ is also able to diffuse between the hexagonal channels in the crystallographic ab plane (Dₐb = (1.9 ± 0.2) × 10⁻¹⁰ m² s⁻¹), despite the walls of these channels appearing impermeable by single-crystal X-ray crystallography and flexible lattice MD simulations. Observation of such unexpected diffusion in the ab plane suggests the presence of defects that enable effective multidimensional CO₂ transport in a metal–organic framework with nominally one-dimensional porosity.

INTRODUCTION

Synthetic porous materials such as metal–organic frameworks (MOFs) show great potential for energy-efficient gas separations. A key advantage of MOFs compared to traditional porous materials, such as carbons and zeolites, is their highly tunable chemistry, porosity, and topology, which allow the targeted design of materials for particular separations. Frameworks in the MOF-74 family, such as M₄(dobdc) (dobdc = 2,5-dioxido-1,4-benzenedicarboxylate; M = Mg, Mn, Fe, Co, Ni, Cu, Zn, Cd)¹⁻⁸ and the expanded variants M₉(dobpdc) (dobpdc = 4,4'-dioxidobiphenyl-3,3'-dicarboxylate; M = Mg, Mn, Fe, Co, Ni, Zn),⁹⁻¹¹ in particular demonstrate promise for the separation of CO₂ from simulated fossil fuel flue gas mixtures¹²⁻¹⁴ as well as for CO₂/CH₄ separations.¹⁵ These frameworks feature one-dimensional hexagonal channels lined with a high density of coordinatively unsaturated metal centers (see Figure 1), and their CO₂ capture performance has been found to improve dramatically upon appending diamines to the metal sites. The diamine-appended frameworks cooperatively adsorb CO₂ via the formation of ammonium carbamate chains, resulting in a dramatically improved performance in the presence of water as well as low regeneration energies.⁹,¹⁰,¹⁶⁻²⁰

Despite the promise of MOF-74 materials for a variety of gas separations, an understanding of gas diffusion in this class of materials is lacking. The importance of strong CO₂–metal interactions in dictating local energy barriers for diffusive motion between different binding sites in Mg₄(dobdc) has previously been highlighted.²¹ Nuclear magnetic resonance

Received: September 4, 2017
Published: January 4, 2018

DOI: 10.1021/jacs.7b09453
J. Am. Chem. Soc. 2018, 140, 1663−1673
directionality of gas diffusion within materials of this type. In particular, the magnitude and directionality of gas diffusion within the one-dimensional nanosized pores of these adsorbents are poorly understood, even though such parameters have a substantial impact on their practical utility. By understanding gas diffusion in this class of MOFs, we hope to guide the design of new materials with improved gas transport properties.

Pulsed-field gradient (PFG) NMR spectroscopy allows the measurement of self-diffusion coefficients of pore-confined molecules\textsuperscript{24–30} and uses pulses of magnetic field gradients to encode and decode the positions of the molecules. In this technique, a series of NMR experiments are first carried out with magnetic field gradient pulses of increasing strength, and self-diffusion coefficients are subsequently obtained by fitting the decay of the NMR signal intensity as a function of the field gradient.\textsuperscript{31} PFG NMR spectroscopy is an attractive means of characterization, as it can also be used to measure diffusion anisotropy.\textsuperscript{31–34} For example, aligning single crystals of a material at different orientations relative to the magnetic field gradient direction (the direction in which diffusion is measured) allows one to probe the diffusion of adsorbed molecules in different crystallographic directions.\textsuperscript{35–37} An alternative approach that does not require oriented single crystals is to make use of residual anisotropic NMR interactions in static polycrystalline samples, wherein the resonance frequency is dependent on the crystallite orientation. Using the latter approach, residual quadrupole interactions (\(^2\)H NMR) combined with PFG experiments have been used to measure the diffusion anisotropy of \(^2\)H\(_2\)O in a polycrystalline liquid crystal sample.\textsuperscript{36} Later studies employed similar approaches based on residual chemical shift anisotropies.\textsuperscript{37–39} In particular, the qualitative analysis of powder line shapes for CO\(_2\) adsorbed in a metal–organic framework can be used to determine the preferred crystallographic direction for self-diffusion.\textsuperscript{39,40}

Here, we utilize the residual chemical shift anisotropy of CO\(_2\) confined in the nanopores of Zn\(_2\)(dobpdc) to extract the diffusion anisotropy as a function of gas pressure. The large single crystals of this material (up to \(\sim 750 \mu m\) in length) facilitate the measurement of diffusion of the pore-confined gas molecules, because exchange with the gas outside of the pores is negligible under the experimental conditions. Our quantitative analysis of the evolution of the entire spectral line shape with increasing pulsed field gradient strength offers precise determination of self-diffusion coefficients parallel and perpendicular to the hexagonal channels. Molecular dynamics (MD) simulations and in situ single-crystal X-ray diffraction measurements are employed as aids in trying to understand the results.

## RESULTS AND DISCUSSION

**NMR Spectra of \(^{13}\)CO\(_2\)-Dosed Zn\(_2\)(dobpdc).** The reaction of Zn(NO\(_3\))\(_2\)-6H\(_2\)O and H\(_2\)dobpdc at 100 °C for 2 days\textsuperscript{17} yielded needle-like crystals of Zn\(_2\)(dobpdc) (Figure 2a, see also Figure 1). Successively washing the crystals with N,N-dimethylacetamide, methanol, and toluene, followed by activation for 12 h at 250 °C under flowing Ar, yielded crystals with lengths of up to 750 \(\mu m\) and widths of up to 120 \(\mu m\) (Figure 2b, Figure S1), which were then used to prepare NMR samples.

![Figure 1. A portion of the crystal structure of Zn\(_2\)(dobpdc) at 298 K shown from two different perspectives, as determined by single-crystal X-ray diffraction. Light blue, red, gray, and white spheres represent Zn, O, C, and H atoms, respectively.](image1)

![Figure 2. (a) Photograph of as-synthesized crystals of Zn\(_2\)(dobpdc) inside a sealed tube. (b) Microscope image of crystals in air following washing and activation.](image2)
The \( \text{Zn}_2 \)(dobpdc) crystals were dosed ex situ with \( ^{13}\text{CO}_2 \) using a custom-built gas dosing manifold (Figure 3a). Static \( ^{13}\text{C} \) NMR spectra of these crystals exhibit a feature that can be assigned to \( ^{13}\text{CO}_2 \) confined within the pores of the framework (Figure 3b). Measurements of the \( ^{13}\text{CO}_2 \) adsorption isotherm and integrated NMR signal intensities from 0 to 2026 mbar of \( ^{13}\text{CO}_2 \) (Figure S2) indicate occupancies of up to 0.9 \( ^{13}\text{CO}_2 \) molecules per metal site at the highest studied pressure. An unusual powder pattern line shape is apparent in the NMR spectra, with the overall width depending on the gas pressure.

Previously reported \( ^{13}\text{C} \) NMR spectra for solid \( \text{CO}_2 \) and our measured spectrum for gaseous \( \text{CO}_2 \) (Figure 3c) provide a useful reference here.\(^{41}\) In the gas phase, isotropic molecular rotation yields a narrow resonance at the isotropic chemical shift, \( \delta_{\text{iso}} = 127.7 \) ppm. In the solid state, \( \text{CO}_2 \) exhibits a significant chemical shift anisotropy (\( \Delta_{\text{CSA}} \), arising from the highly anisotropic shape of the molecule. For solid \( \text{CO}_2 \), \( \Delta_{\text{CSA}} = \delta_{\parallel} - \delta_{\text{iso}} \) is approximately \(-210 \) ppm, where \( \delta_{\text{iso}} = (2\delta_{\perp} + \delta_{\parallel})/3 \), and intensity at \( \delta_{\parallel} \) arises from \( \text{CO}_2 \) molecules that are oriented perpendicular to the applied magnetic field \( (B_0) \), while intensity...
at $\delta_\parallel$ arises from CO$_2$ molecules oriented parallel to the applied field.

Pore-confined CO$_2$ in Zn$_2$(dobpdc) crystals exhibits a residual chemical shift anisotropy. In this regime, crystals with different orientations relative to $B_0$ exhibit different chemical shifts for confined CO$_2$ molecules. This effect arises because the molecular reorientation inside the pores is not isotropic, and CO$_2$ has preferred average orientations relative to the crystal frame (Figure 3b). We stress that it is the different orientations of the crystals relative to $B_0$ that give rise to the observed spread of chemical shifts in Figure 3b. While the metal is the primary binding site for CO$_2$ in these materials, CO$_2$ undergoes translations between different metal sites on the NMR time scale, and a symmetric narrow line is thus expected for a single crystal (reflecting the average environment of pore-confined CO$_2$ on a millisecond time scale; see below for more details). We note that similar line shapes for confined CO$_2$ have previously been observed in other (noncubic) MOFs including the isoreticular framework Mg$_2$(dobdc). We hypothesize that intensity at the left-hand edge of the spectrum arises from CO$_2$ inside the pores of crystals that are oriented perpendicular to the magnetic field ($B_0$) direction and intensity at the right-hand edge arises from CO$_2$ in the pores of crystals that are oriented parallel to the field direction (Figure 3b). With this assumption, the calculated $\Delta_{CSA}$ ($= \delta_\parallel - \delta_\perp$) values are negative. The observed line shapes deviate significantly from that expected for a randomly oriented polycrystalline powder sample (shown as a dashed blue line in Figure 3b). This deviation arises because the large crystal sizes result in a partial macroscopic alignment of their long axis with the long axis of the NMR tube, leading to additional intensity at $\delta_\parallel$. Consistent with this observation, we find that physical shaking of the crystal samples results in marked changes of intensity in different regions of the spectra (Figure S3).

Our assignments of $\delta_\perp$ and $\delta_\parallel$ are supported by four pieces of evidence. First, magic angle spinning (MAS) spectra collected for gas-dosed single crystals (Figure S4) yield a negative chemical shift anisotropy, as well as an isotropic chemical shift consistent with our assignments. Second, measurements of...
Zn$_2$(dobpdc) powder dosed with $^{13}$CO$_2$ (Figure 3d, see Experimental Section for powder synthesis) yield a narrow resonance with an isotropic chemical shift consistent with our assignments (see Figure 3e). Interestingly, a residual chemical shift anisotropy is not visible in this spectrum, suggestive of fast chemical exchange between different crystallites (typical crystallite lengths are <15 μm, and widths are <2 μm; see Figure S5) or that the preferred orientation of CO$_2$ relative to the crystal frame is slightly different in the powder and the crystal samples, which were synthesized via different routes. Third, the sign of the assigned $\Delta_{\text{CSA}}$ agrees with that calculated by MD simulations (see below). Finally, the diffusion behavior observed below is consistent with our assignments, with $D_\parallel$ ≫ $D_\perp$ as anticipated. An alternative assignment of the spectra in Figure 3b would be that intensity at the left-hand edge arises from crystals that are parallel to $B_0$ and that intensity at the right-hand edge arises from crystals that are perpendicular to $B_0$ (i.e., positive $\Delta_{\text{CSA}}$). However, we reject this alternative assignment, as it is inconsistent with the four observations above.

The graphs in Figure 3e show the variations of $\Delta_{\text{CSA}}$ and $\delta_{\text{iso}}$ with $^{13}$CO$_2$ pressure. These data are in agreement within error for two independent samples measured at different field strengths, which we call “sample 1” and “sample 2” (see Figure S6 for spectra of sample 2, collected with in situ dosing). Notably, $\delta_{\text{iso}}$ shows small variations at lower pressures and then appears to plateau at higher pressures, while the magnitude of $\Delta_{\text{CSA}}$ increases continuously with pressure (see below for discussion of this).

**Anisotropic Diffusion of CO$_2$.** PFG NMR measurements of sample 1 dosed ex situ with 625 mbar of $^{13}$CO$_2$ are shown in Figure 4a. In these measurements, magnetic field gradient pulses were applied along the laboratory z-direction (i.e., parallel to $B_0$; see Figure S7). Upon application of pulsed magnetic field gradients of increasing magnitude, the right-hand edge of the spectrum decays more rapidly than the left-hand edge. Face-indexing of a single crystal confirmed that the c-axis (and therefore the one-dimensional pores) runs along the long axis of the needle-like Zn$_2$(dobpdc) crystals. Using this observation and our prior assignments of the spectra (Figure 3b), we can obtain CO$_2$ self-diffusion coefficients both parallel ($D_\parallel$) and perpendicular ($D_\perp$) to the hexagonal channels of Zn$_2$(dobpdc).

The observed chemical shift, $\delta$, for pore-confined CO$_2$ in a crystal oriented at an angle (θ) relative to $B_0$ is given by

$$\delta(\theta) = \delta_{\text{iso}} + \Delta_{\text{CSA}} \left( \frac{3 \cos^2 \theta - 1}{2} \right)$$

The effective self-diffusion coefficient ($D$) measured using magnetic field gradients in the z-direction depends on $\theta$ in a similar way and is given by

$$D(\theta) = D_\parallel \cos^2 \theta + D_\perp \sin^2 \theta$$

Finally, the decay of signal intensity ($I$), normalized by the intensity for zero gradient ($I_0$), is given by

$$\frac{I}{I_0} = \exp[-bD(\theta)]$$

The constant $b$ is proportional to the square of the gradient strength ($g$) and also depends on the gyromagnetic ratio ($\gamma$) of the studied nucleus and a number of experimental parameters (see Figure S8 for definitions of $\Delta$, $\delta$, $\tau$) and is given by

$$b = g^2(2\delta \gamma)^2 \left( \frac{\Delta - \delta - \tau}{\phi} \right)$$

The self-diffusion coefficients $D_\parallel$ and $D_\perp$ are therefore the only free parameters in the fit to the data in Figure 4b using eq 3. Here, we fit against all of the data (i.e., $I/I_0$ vs $b$ across all crystal orientations, $\theta$) simultaneously to obtain diffusion coefficients with relatively small uncertainties. This analysis method is preferred to individual fits of integrated signal intensities across a small range of chemical shifts, which revealed similar results but with errors at least three times larger due to uncertainty in the fits. The dominant source of error in our analysis is most likely systematic error arising from the finite peak line widths that arise from homogeneous broadening, such that for a given $\delta$, crystals at a range of $\theta$ values contribute to the signal $^{13,37}$ Indeed, this is evidenced by an apparent narrowing of the signal at $\delta_2$ at the largest applied gradient strengths (Figure 4a). This arises from the signal decay for CO$_2$ in crystals with orientations, $\theta$, close to but unequal to 90° $^{39}$ We note that it would be interesting to carry out PFG NMR with different gradient orientations. $^{39}$ Under such conditions the form of the PFG NMR decays would differ and could further improve the accuracy of our diffusion measurements.

A global fit to eq 3 (Figure 4b) affords $D_\parallel = (5.8 \pm 0.1) \times 10^{-9}$ m$^2$ s$^{-1}$ and $D_\perp = (1.9 \pm 0.2) \times 10^{-10}$ m$^2$ s$^{-1}$. These values suggest that the self-diffusion of CO$_2$ is ~30 times faster along the hexagonal channels compared to diffusion between the channels at this pressure. It is surprising that gas diffusion perpendicular to the channels is possible at all, given the apparent lack of pore windows between neighboring channels in the $ab$ plane, as judged from the crystal structure (see below). We note that the measured values are of similar order to self-diffusion values of bulk liquids, though they are more than 3 orders of magnitude smaller than the self-diffusion coefficient of gaseous CO$_2$ ($1.1 \times 10^{-5}$ m$^2$ s$^{-1}$ at 25 °C, 1 bar) $^{40}$ Our measured values are of similar order to the diffusion of CO$_2$ in other large-pore MOFs, such as HKUST-1 ($D = 1.7 \times 10^{-9}$ m$^2$ s$^{-1}$) $^{49}$ Zn$_2$(bdc)$_2$(dabco) ($D = 1.1 \times 10^{-8}$ m$^2$ s$^{-1}$ and $D_\perp = 3.7 \times 10^{-9}$ m$^2$ s$^{-1}$) $^{39}$ and MIL-47(V) ($D = 10^{-8}$ – $10^{-9}$ m$^2$ s$^{-1}$, depending on loading level). $^{50}$

Because these PFG NMR experiments probe diffusion on length scales on the order of tens of micrometers, $D_\parallel$ is the effective self-diffusion coefficient as CO$_2$ moves between thousands of pores, rather than within a single pore (a length scale that is inaccessible by this technique). Furthermore, for the observation times ($\Delta$) studied here, the experiments probe diffusion in the pore-confined phase. Indeed, the root-mean-square displacements (RMSDs) of confined CO$_2$ in our experiments are smaller than the typical crystal dimensions (Figure 2b, Figure S1). The RMSD is calculated using the expression ($n \Delta D)^{0.5}$ ($n = 2$ for diffusion in the c-direction and $n = 4$ for diffusion in the $ab$ plane). For example, given $\Delta = 80$ ms, the RMSDs are 30 μm along the c axis, and 8 μm in the $ab$ plane, values that are smaller than the typical crystal lengths (hundreds of micrometers) and widths (tens of micrometers), respectively (Figure S1). Furthermore, diffusion coefficients measured for different $\Delta$ in the range of 20–160 ms show minimal variation, indicating that exchange of pore-confined CO$_2$ with CO$_2$ outside of the pores is not significant at these time scales (Figure S9). Exchange spectroscopy additionally
reveals that significant exchange of $^{13}$CO$_2$ between crystals occurs on time scales of hundreds of milliseconds, much longer than the time scales employed here (Figure S10). In contrast, PFG NMR measurements on the powdered sample (typical crystallite lengths up to 15 μm, widths up to 2 μm; Figure S5) yield much larger effective self-diffusion values, on the order of $10^{-7}$–$10^{-8}$ m$^2$ s$^{-1}$, reflecting the more rapid exchange of CO$_2$ between different crystallites as well as rapid diffusion in the intercrystallite space. Indeed, the probed RMSDs in this experiment are between $\sim$100–300 μm for a relatively short observation time of 20 ms, highlighting the need for large crystals to reliably measure intracrystalline diffusion coefficients.

Further PFG NMR experiments on samples dosed at different $^{13}$CO$_2$ gas pressures yielded similar results (see PFG NMR spectra and fits in Figures S11 and S12). The measured diffusion parameters at pressures of 635, 1010, and 2026 mbar are shown in Figure 4c and Table 1. The diffusion values appear relatively constant at these pressures, a result that is perhaps unsurprising given the relatively small variations in CO$_2$ loading in the pressure range investigated (see CO$_2$ adsorption isotherm in Figure S2a).

**Molecular Dynamics Simulations.** To further probe CO$_2$ diffusion in Zn$_2$(dobpdc), we performed MD simulations using (i) a fully flexible MOF lattice model and (ii) a rigid MOF lattice. Figure 5a shows the time-evolution of the mean square displacement (MSD) of CO$_2$ molecules in the different crystallographic directions for the flexible lattice model with a CO$_2$ pressure of 1 bar. Since log(MSD) is given by log(nD) + log(t) in the diffusive regime, a slope of 1 and a y-intercept of log(nD) are anticipated on a log–log plot, provided that sufficiently long times are probed to reach this regime. Inspection of the plot in Figure 5a thus informs an assessment of the diffusion behavior of CO$_2$ in Zn$_2$(dobpdc).

Long-range self-diffusion of CO$_2$ occurs along the $c$ axis, as anticipated, with the diffusive regime (slope = 1) being reached at times longer than $\sim$10 ps (Figure 5a). In contrast, diffusion of CO$_2$ in the ab plane is limited to displacements within a single pore, with no diffusive jumps occurring between different channels. This observation is reflected by the apparent plateauing of the MSD in the ab plane at long times, with the diffusive regime never being reached. At 298 K, the MSD climbs to $\sim$160 Å$^2$ at a time of 10 ns, corresponding to a RMSD of $\sim$13 Å, which, as expected, is less than the pore diameter of $\sim$22 Å.

Similar behavior is apparent at all three studied temperatures of 298, 400, and 500 K, as well as at a lower pressure of 0.01 bar (Figure S13a), indicating the absence of a thermally activated diffusion mechanism that could allow CO$_2$ to move between pores. MD simulations with a rigid lattice also revealed the absence of long-range diffusion in the ab plane. This observation from MD simulations, which is based upon the assumption of perfect crystals, is in contrast to the results from PFG NMR experiments, where crystal defects may affect CO$_2$ diffusion (see below).

**Table 1. Self-Diffusion Coefficients of $^{13}$CO$_2$ in Zn$_2$(dobpdc) at 298 K**

<table>
<thead>
<tr>
<th>Pressure, mbar</th>
<th>$D_{ab}$ m$^2$ s$^{-1}$</th>
<th>$D_{c}$ m$^2$ s$^{-1}$</th>
<th>$D_{ab}$/$D_{c}$</th>
</tr>
</thead>
<tbody>
<tr>
<td>635</td>
<td>5.8(±0.1) × 10$^{-9}$</td>
<td>1.9(±0.2) × 10$^{-10}$</td>
<td>30 ± 3</td>
</tr>
<tr>
<td>1010</td>
<td>6.2(±0.2) × 10$^{-9}$</td>
<td>2.3(±0.3) × 10$^{-10}$</td>
<td>27 ± 4</td>
</tr>
<tr>
<td>2026</td>
<td>6.3(±0.2) × 10$^{-9}$</td>
<td>1.4(±0.2) × 10$^{-10}$</td>
<td>48 ± 7</td>
</tr>
</tbody>
</table>

The self-diffusion coefficients along the $c$ axis ($D_{c}$) determined from MD simulations are shown in Figure Sb and Figure S3b. These simulations allow access to pressures from 10 mbar to 100 bar (a much wider range of pressures, and therefore CO$_2$ loadings, than is accessible with our current NMR apparatus) and reveal loading-dependent diffusion behavior. A clear increase in $D_{c}$ is observed at very low loadings, followed by decreases at higher loadings. We postulate that at the lowest loadings, the small $D_{ab}$ values result from the high density of open metal sites. Indeed, DFT calculations of CO$_2$ diffusion pathways in a similar MOF, Mg$_2$(dobdc), have shown that the energy barrier for CO$_2$ to move from a metal site to a neighboring unoccupied metal site is $\sim$300 meV ($\sim$30 kJ mol$^{-1}$).$^{21}$ After some of the metal sites become occupied by CO$_2$, additional lower energy pathways may give rise to the observed increase in $D_{ab}$, particularly diffusive motion of CO$_2$ through the pore center, which bypasses interaction with metal sites that are already occupied by other CO$_2$ molecules (for Mg$_2$(dobdc), a barrier $<40$ meV or $>4$ kJ mol$^{-1}$).$^{21}$
availability of these additional pathways is exemplified by the upturn in $D_{\perp}$ between 750 and 1000 mbar, which also corresponds to the point at which a loading of 1 CO$_2$ per Zn$^{2+}$ site is surpassed in the simulations. At higher CO$_2$ loadings, the observed decreases in $D_{\parallel}$ are due to CO$_2$–CO$_2$ interactions. We note that PFG NMR gives smaller self-diffusion values than MD by a factor of $\sim$10, with such discrepancies not uncommon in the literature.$^{51,52}$ In the present case, differences in the experimental and simulated values may arise from both the presence of crystal defects in the experiment and small errors in the parametrization of the force field used for simulations.

The MD simulations also allow a prediction of the chemical shift anisotropy parameter, $\Delta_{\text{CSA}}$. We assume that the observed chemical shift is dependent only on the molecular orientation of CO$_2$ relative to the applied magnetic field and that framework–CO$_2$ interactions do not significantly perturb the chemical shift. This is a reasonable assumption given that our experimentally observed isotropic chemical shifts do not deviate significantly from that observed for free CO$_2$ gas (Figure 3e). Furthermore, we assume that CO$_2$ molecules exchange rapidly between different adsorption sites within the pores, such as primary sites (metal-bound) and secondary sites (not directly metal-bound) on the NMR time scale. Following a previously published procedure,$^{60,61}$ we first obtain probability distributions for the molecular orientations of CO$_2$ molecules in the crystal frame from the MD trajectories. We then obtain motionally averaged chemical shift tensors by performing a weighted average using the orientation probability distributions (see Supporting Information for details). Predicted $\Delta_{\text{CSA}}$ values are then obtained from these tensors and are compared to the reported values for solid CO$_2$.$^{41}$ $\Delta_{\text{CSA,solid}}$ as the order parameter $\Delta_{\text{CSA}}/\Delta_{\text{CSA,solid}}$ (Table 2).

Consistent with the NMR experiments, negative $\Delta_{\text{CSA}}$ values are obtained that increase in magnitude with pressure (Table 2). The values show reasonable agreement, with the discrepancies similar to those in a related study.$^{40}$ The values from MD are systematically larger in magnitude than the NMR values. This disagreement likely arises from the presence of crystal defects that will affect the orientations of the CO$_2$ molecules in the pores. Our observed increase of the magnitude of $\Delta_{\text{CSA}}$ with pressure is consistent with recent NMR measurements on the related metal–organic framework Mg$_2$(dobdc).$^{23}$ The observed increases in $\Delta_{\text{CSA}}$ with pressure likely reflect a competition between two factors: (i) the preferred orientation of CO$_2$ strongly interacting with Zn$^{2+}$ in primary adsorption sites and (ii) the occupation of secondary binding sites with a range of different orientations. At low pressures, the majority of CO$_2$ is bound in primary binding sites (see X-ray measurements below) and factor i will tend to dominate. At higher CO$_2$ pressures, an increasingly large population of CO$_2$ in secondary sites is present in the pores, and factor ii becomes increasingly important.

**X-ray Crystallography.** To further explore the observed CO$_2$ diffusion behavior and the nonzero value of $D_{\parallel}$ from PFG NMR, we used in situ single-crystal X-ray diffraction to investigate the structure of Zn$_2$(dobpdc) before (Figures 1 and S14) and after (Figure 6) exposure to 1.01 bar CO$_2$ at 298 K. Activated Zn$_2$(dobpdc) is isostructural with previously reported solvated structures (Figure 1),$^{5,17}$ and no clear distortions of the framework are evident following the adsorption of CO$_2$ (Figure 6).

![Figure 6](image_url)

**Table 2.** $\Delta_{\text{CSA}}$ Values for $^{13}$CO$_2$ in Zn$_2$(dobpdc) at 298 K from MD Simulations and NMR Spectroscopy (Sample 1)

<table>
<thead>
<tr>
<th>method</th>
<th>pressure, mbar</th>
<th>$\Delta_{\text{CSA}}$, ppm</th>
<th>$\Delta_{\text{CSA}}/\Delta_{\text{CSA,solid}}$</th>
</tr>
</thead>
<tbody>
<tr>
<td>MD</td>
<td>750</td>
<td>−51.4</td>
<td>0.24</td>
</tr>
<tr>
<td></td>
<td>1000</td>
<td>−51.5</td>
<td>0.25</td>
</tr>
<tr>
<td></td>
<td>7500</td>
<td>−54.9</td>
<td>0.26</td>
</tr>
<tr>
<td>NMR</td>
<td>625</td>
<td>−6.5</td>
<td>0.03</td>
</tr>
<tr>
<td></td>
<td>1010</td>
<td>−8.7</td>
<td>0.04</td>
</tr>
<tr>
<td></td>
<td>2026</td>
<td>−12.1</td>
<td>0.06</td>
</tr>
</tbody>
</table>

Figure 6. Views of the structure of Zn$_2$(dobpdc) under 1.01 bar CO$_2$ obtained from single-crystal X-ray diffraction at 298 K. Light blue, red, gray, and white ellipsoids represent Zn, O, C, and H atoms, respectively. Note that CO$_2$ was found to be disordered over two orientations with occupancies of 34% and 30%. Thermal ellipsoids are drawn at a 50% probability level. In panel b, CO$_2$ atoms are omitted to enable a clear view of the pore walls.
We further examined the framework structure to determine if CO₂ can pass through the small space between dobpdc⁻⁴⁻ linkers in the ab plane (Figure 6b). The H⋯H separations between adjacent linkers are 2.5436(3) Å (central H atoms) and 2.7781(3) Å (outer H atoms). Taking the H atom van der Waals radius to be 1.1 Å, the maximum pore size of ~0.6 Å is much smaller than the diameter of a CO₂ molecule (estimated as 3.4 Å, twice the van der Waals radius of carbon), such that diffusion of CO₂ in the ab plane is not expected. There are thus two most likely causes of the unexpected diffusion in the ab plane: (i) dynamics or (ii) defects.

Thermal ellipsoids drawn at a 50% probability level are shown in Figure 6 to aid the visualization of framework dynamics. The ellipsoids of the framework atoms are relatively small under the experimental conditions at 298 K, suggesting that there is little movement of the framework atoms over the average of all unit cells. We note that coordination of three oxygen donors at each end of the dobpdc⁻⁴⁻ to the metal chains (one bridging and one nonbridging carboxylate oxygen and one bridging arylxide oxygen) restrain the linker, limiting the motion of the phenyl rings. This contrasts to other common MOF linkers such as bpdc⁻⁻ (bpdc⁻⁻ = biphenyl-4,4’-dicarboxylate) that only bind through carboxylate groups. In these frameworks, the phenyl groups of the linker are often free to rotate about the phenyl-carboxylate C—C bond, which results in large thermal ellipsoids even at 100 K. Note that our crystallographic data does not rule out an activated process in which the framework distorts to allow CO₂ to pass through a transient pore, which would not be discernible by crystallography. However, the flexible lattice MD simulations (Figure 5a) reveal no evidence for such an activated process, even at temperatures as high as 500 K. We therefore conclude that framework dynamics are unlikely to be responsible for the nonzero D₂ value observed by PFG NMR spectroscopy.

A second explanation for diffusion between channels in the ab plane is the presence of crystal defects. For example, missing linkers or missing Zn²⁺ ions could result in porosity in the ab plane that would otherwise be absent. Refinement of the occupancy of the Zn²⁺ site relative to a fixed occupancy of the dobpdc⁻⁴⁻ linkers suggests a slight deficiency of zinc, with 94.8(4)% and 96.8(4)% of the anticipated electron density observed for activated and CO₂-dosed (1.01 bar) Zn₂(dobpdc), respectively. Inductively coupled plasma optical emission spectrometry (ICP-OES) also indicated a slight deficiency of zinc compared to the expected molecular formula. Although elemental analysis gave H and C mass ratios that agree with the expected values for Zn₂(dobpdc) within error (Table S2), we note that some types of defects may not lead to significant changes in the material stoichiometry. The nature of the possible defects in this class of frameworks remains an active area of investigation.

Returning to the X-ray diffraction experiments, the CO₂-loaded structure (Figure 6) allows two preferred binding sites for CO₂ to be resolved. The primary adsorption site lies at the open Zn²⁺ coordination sites of the framework. We find that the Zn—O_CO₂ distance is 2.528(12) Å, which is slightly longer than the distance of 2.43(4) Å determined by powder neutron diffraction from an earlier study of CO₂-dosed Zn₂(dobpdc) at 10 K. This disparity in the distances may arise from the large difference in data collection temperature for the two structures. By comparison, the most common Zn—O_CO₂ distance from our MD simulations at similar conditions (1 bar CO₂, 298 K) is 2.8—2.9 Å. This is in reasonable agreement with the DFT calculations to which the MD force field was parametrized, though slightly longer than that from X-ray diffraction. From previous DFT calculations using a dispersion-corrected functional, it was observed that the preferred Zn—O_CO₂ distance in Zn₂(dobpdc) is 2.7 Å, which is approximately 0.2 Å longer than the other equilibrium M—O_CO₂ distances calculated (M = Mg, Co, Fe, and Ni) and also slightly longer than that observed experimentally by diffraction. A systematic overestimation of the M—O_CO₂ distances from DFT calculations has been observed previously with the same dispersion-corrected functional. The longer distance obtained from MD may also be partly accounted for by the relatively large thermal ellipsoid of the metal-bound oxygen in the X-ray diffraction structure (Figure 6a), which will tend to lengthen the Zn—O_CO₂ distance.

The bound CO₂ molecule in our X-ray structure exhibits disorder over two orientations with Zn—O—C angles of 152.3° and 122.3° and occupancies of 34% and 30%, respectively (Figure 6a). These occupancies correspond to a total of 0.64 CO₂ molecules per Zn, which is similar to the value of 0.72 expected from the gas sorption isotherm (Figure S2). The discrepancy between the two values likely arises due to the presence of some CO₂ at secondary binding sites, which are not resolvable in the diffraction experiments carried out here but can be seen in our MD simulations (Figure S15). In the future, variable-pressure crystallographic experiments may help us to further understand the nature of the different binding sites, as has been performed for the smaller pore M₁₂(dobpdc) materials.

CONCLUSIONS

The foregoing results demonstrate how the residual chemical shift anisotropy for CO₂ confined in the pores of the metal–organic framework Zn₂(dobpdc) allows measurement of the diffusion anisotropy using PFG NMR spectroscopy. Surprisingly, we find that the pore-confined gas is able to diffuse between adjacent framework channels in the ab plane, with self-diffusion coefficients of D₂ = (5.8 ± 0.1) × 10⁻⁹ m² s⁻¹ and D₁ = (1.9 ± 0.2) × 10⁻¹⁰ m² s⁻¹ at 298 K and a pressure of 625 mbar of CO₂. Measurements at gas pressures between 625 and 2026 mbar yielded very similar diffusion values, while MD simulations showed significant variations of self-diffusion at more extreme pressures. In situ single-crystal X-ray diffraction experiments revealed no obvious structural distortions upon the adsorption of CO₂, while flexible lattice MD simulations revealed no diffusion between different channels in the ab plane. These findings indicate that defects in the framework structure are likely responsible for the observed nonzero diffusion in the ab plane. The marked diffusion anisotropy opens considerable opportunities for crystal engineering for CO₂ capture as well as other gas storage and separation applications, both for adsorbents and mixed-matrix membranes. For example, one could tune the crystal shapes and sizes to optimize fast adsorption and desorption, while the possibility of orienting MOF crystals with anisotropic diffusion properties in membranes should also be explored. The approaches described herein should be readily applicable to other porous materials exhibiting anisotropic pore structures.

EXPERIMENTAL SECTION

Materials and Methods. All manipulations were performed under an N₂ atmosphere in a VAC atmospheres glovebox or using standard...
in space group P3\(_{1}\)21 as inversion twins based on Flack parameter values near 0.5. Raw data were corrected for Lorenz and polarization effects using Bruker AXS SAINT software\(^2\) and were corrected for absorption using SADABS.\(^3\) The structures were solved using SHELXT\(^4\) and refined using SHELXL\(^5\) operated in the OLEX2\(^6\) interface. Thermal parameters were refined anisotropically for all non-hydrogen atoms. All hydrogen atoms were placed geometrically and refined using a riding model. Disorder modeling of the CO\(_2\) molecules bound to Zn\(^{2+}\) in Zn\(_2\)(dobpdc)·1.26CO\(_2\) required the use of displacement parameter (RIGU and SIMU) and distance (DFIX) restraints.

Face-indexing was performed at the ChaXray Facility at UC Berkeley with a hexanes-solvated single crystal of Zn\(_2\)(dobpdc) exposed to air. The crystal was mounted on a Bruker QUAZAR diffractometer (Mo K\(_\alpha\), \(\lambda = 0.71073\) Å) equipped with a Bruker APEX II detector. Analysis of the crystal was performed using the Bruker APEX3 software.\(^6\)

Preparation of Samples for NMR Experiments. Valved high-pressure 5 mm NMR tubes (Norell) were filled with activated crystals, and the tube was connected to a custom dosing apparatus (Figure 2a). Samples were first evacuated for at least 30 min before dosing with \(^{13}\)C-enriched CO\(_2\) gas (Sigma-Aldrich, 99 atom % \(^{13}\)C, <3 atom % \(^{15}\)O) for 30 min for equilibration. For \(\textit{ex situ}\) dosing (as performed for sample 1), the valve on the NMR tube was then closed before NMR measurements. For \(\textit{in situ}\) dosing (sample 2), measurements were made with the sample still attached to the custom dosing manifold. Here a long piece of copper tubing was used to connect the sample inside the NMR magnet to the dosing manifold.

NMR Experiments. Static NMR experiments were conducted using the magnetic field strengths indicated in the figure captions. Liquid state probes and 5 mm sample tubes were used for all static measurements. MAS experiments (spectra are shown in the Supporting Information) were conducted using a DOTY 4 mm double resonance probe. For PFG NMR experiments, a Diff-30 probe and a 13-1interval pulse sequence with bipolar magnetic field gradient pulses was employed (Figure S8).\(^7\) Gradients were calibrated using the known self-diffusion coefficient of water at 25 °C (2.3 × 10\(^{-9}\) m\(^2\) s\(^{-1}\)) to test the system performance, diffusion coefficients were measured and verified against literature values for acetone, cyclohexane, toluene, and glycerol at 24 °C,\(^68–70\) covering a range of self-diffusion coefficients from 5 × 10\(^{-9}\) m\(^2\) s\(^{-1}\) (acetone) to 2 × 10\(^{-12}\) m\(^2\) s\(^{-1}\) (glycerol). For experiments on MOF samples, the effective duration of the individual magnetic field gradient pulses (\(\delta\)) was 1 ms, and the maximum magnetic field gradient strength applied was 9 T m\(^{-1}\), with sine-shaped gradient pulses used. Observation times (\(\Delta\)) are stated in the text and were set to integer multiples of the period (20 ms) of the 50 Hz mains alternating current to mitigate potential issues from residual mains hum.\(^61\) The parameter \(\tau\) was 3 ms (Figure S8). The PFG NMR decays were analyzed using eqs 3 and 4 and a fitting program written in Matlab software.\(^6\)\(^2\)\(^3\)\(^\mathrm{C}\) NMR chemical shifts are referenced to the ethylene carbon of neat ethanol at 57.4 ppm as a secondary reference relative to tetramethylsilane (1 vol % in deuterated chloroform) at 0.0 ppm, except for MAS experiments that are referenced to the tertiary carbon atom in adamantane at 38.5 ppm.

Molecular Dynamics Simulations. Molecular dynamics (MD) simulations were performed in the NVT ensemble with LAMMPS\(^1\) using a time step of 1.0 fs and the Nosé–Hoover chain thermostat.\(^72,73\) The dynamics of CO\(_2\) were treated using a rigid-body time integrator.\(^74\) The CO\(_2\) molecules were modeled using the TraPPE force field\(^75\) and CO\(_2\)–framework interactions were modeled using a previously published force field.\(^76\) The flexible lattice simulations required a molecular model for the framework atoms. Bond, angle, dihedral, and torsion parameters for linker molecules were taken from the consistent valence force field,\(^77\) while Zn\(^{2+}\) ions were modeled using a cationic dummy model.\(^78\) Lorentz–Berthelot mixing rules were used to calculate cross-interactions between Zn\(^{2+}\) and linker atoms. For simulations at different pressures, the framework was loaded with a number of CO\(_2\) molecules corresponding to previously published isotherms.\(^58\) All MD simulations were equilibrated for 1 ns, and
production runs lasted at least 10 ns to ensure that the mean-squared displacement (MSD) became linear as a function of time. The order-r algorithm \(^6,^7,^9\) was used to calculate the MSD as a function of time, and the 1D self-diffusion coefficients were obtained by fitting the slope of the linear regime: \(D = \frac{1}{6} \lim_{t \to \infty} \frac{1}{t} \langle (r(t) - r(0))^2 \rangle\)

**ASSOCIATED CONTENT**

Supporting Information
The Supporting Information is available free of charge on the ACS Publications website at DOI: 10.1021/jacs.7b09453.

Gas sorption, microscopy, elemental analysis, ICP, additional NMR spectra, characterization of Zn\(_2\)(dobpdc) powder, and additional experimental details (PDF)

Crystallographic data for Zn\(_2\)(dobpdc) (CIF)

Crystallographic data for Zn\(_2\)(dobpdc)-1.26CO\(_2\) (CIF)

**AUTHOR INFORMATION**

Corresponding Authors
*reimer@berkeley.edu
*jrlong@berkeley.edu

ORCID

Alexander C. Forse: 0000-0001-9592-9821

Velencia J. Witherspoon: 0000-0002-2718-6605

Sudi Jawahery: 0000-0001-9950-6199

Berend Smit: 0000-0003-4635-8562

Jeffrey R. Long: 0000-0002-5324-1321

**Notes**

The authors declare no competing financial interest.

**ACKNOWLEDGMENTS**

This research was supported through the Center for Gas Separations Relevant to Clean Energy Technologies, an Energy Frontier Research Center funded by the U.S. Department of Energy (DOE), Office of Science, Office of Basic Energy Sciences, under Award DE-SC0001015. We thank the Philomathia Foundation and Berkeley Energy and Climate Institute for support of A.C.F. through a postdoctoral fellowship, the National Institutes of Health for support of P.J.M. through a postdoctoral fellowship (GM120799), and the Miller Institute for Basic Research in Science for support of J.D.M. through a postdoctoral fellowship. This work employed resources at the Advanced Light Source at Lawrence Berkeley National Lab, a user facility supported by the Director, Office of Science, Office of Basic Energy Sciences, of the DOE under Contract No. DE-AC02-05CH11231. Single-crystal X-ray diffraction equipment within the UC Berkeley CheXray Facility was supported by NIH shared Instrumentation Grant S10-RR027172. We thank ACaNet, the Aachen-California Network of Academic Exchange (DAAD Germany) for supporting a research visit of A.C.F. to RWTH Aachen University. We further thank Stefan Benders and Patrick Offer for useful discussions and assistance with PFG NMR experiments, Kristen Colwell for obtaining SEM images, Julia Oktawiec for assistance with gas adsorption measurements, Richard Bounds for assistance with the development of gas dosing equipment for in situ NMR experiments, and Katie R. Meihaus for providing editorial assistance.

**REFERENCES**


(39) Peksa, M.; Lang, J.; Stallmach, F. Microporous Mesoporous Mater. 2015, 205, 11.
(63) Sheldrick, G. M. SADABS, program for empirical absorption correction of area detector data; University of Göttingen, Germany.